# **Halogen-induced charge transfer polymerization of pyrrole in aqueous media**

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Simultaneous polymerization and doping of pyrrole have been carried out in the presence of a halogenic electron acceptor, bromine  $(Br_2)$  or iodine  $(I_2)$ , in aqueous dispersion or in a two-phase solvent system. The morphology of the polypyrrole (PPY) so produced is granular and porous. The electrical conductivity of **the**  PPY-I<sub>2</sub> charge transfer (CT) complex is of the order of 10<sup>1</sup> ohm<sup>-1</sup> cm<sup>-1</sup> while that of the PPY-Br<sub>2</sub> complex is about one order of magnitude less. Both complexes are stable in the atmosphere. The physicochemical properties of the PPY-I, and PPY-Br, CT complexes prepared under various experimental conditions are examined in detail.

**(Keywords: polypyrrole; charge transfer polymerization; halogen doping; electrical conductivity; thermal analysis)** 

# INTRODUCTION

The report on the exceptional electrical properties of doped polyacetylene in  $1977<sup>1</sup>$  has contributed to a rapidly increasing interest in conducting conjugated polymers during the past few years. Intensive studies have been made on conjugated organic polymers such as  $poly(p\text{-phenylene})^2$ , poly(phenylene sulphide)<sup>3</sup>, poly(diacetylene)<sup>4</sup>, polythiophene<sup>5</sup>, polypyrrole<sup>6</sup> and their derivatives. Indeed, conducting polymers have emerged as a new class of electrical/electronic materials<sup>7</sup>. Of these polymers, the heterocyclic polymers are usually prepared in the form of highly conducting polyheterocyclic cations in the process of electrochemical polymerization $8-10$ . In particular, highly conductive polypyrrole (PPY) film obtained via the electrochemical oxidation and polymerization of pyrrole and Nsubstituted pyrrole derivatives in the presence of various counteranions and electrolytes have been under intensive investigation<sup>8,11-13</sup>. The availability of conductive pyrrole polymers in film form from the electrochemical preparation has encouraged us to search for a chemical method of preparation.

Previous attempts by other investigators at the chemical preparation of pyrrole polymers have resulted mainly in insulating materials<sup>14-16</sup>. Room-temperature conductivity of the PPY films obtained from the typical chemical oxidation of pyrrole is of the order of  $10^{-11}$  ohm<sup>-1</sup> cm<sup>-1</sup>.<sup>16</sup> Moreover, these films are not completely stable in air even at room temperature<sup>16</sup>. These initially insulating films can be doped by bromine  $(Br<sub>2</sub>)$  and iodine  $(I<sub>2</sub>)$  to achieve stable conductivity values of the order of  $10^{-5}$  ohm<sup>-1</sup> cm<sup>-1</sup>.<sup>16</sup> Very recently, Bocchi et al.<sup>17</sup> have reported on the chemical polymerization of conducting polypyrrole in aqueous  $FeCl<sub>3</sub>$  solution and by various oxidizing agents, such as halogens and other metal salts. In this paper, we report on a relatively simple method for the simultaneous polymerization and doping of pyrrole in the presence of a halogen, such at  $Br_2$  or  $I_2$ . The resulting PPY-I<sub>2</sub> and  $PPY-Br<sub>2</sub>$  charge transfer (CT) complexes are quite

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conductive and are completely stable in ambient air. The morphology and physicochemical properties of these complexes prepared under various conditions are examined by means of scanning electron microscopy (SEM), u.v.-visible absorption and infra-red (i.r.) absorption spectroscopy, differential scanning calorimetry (d.s.c.), chemical analysis, thermogravimetric analysis (t.g.a.), conductivity ( $\sigma$ ) and density measurements.

# EXPERIMENTAL

#### *Polymer synthesis and doping*

The monomer, pyrrole (Merck, reagent grade), used for the present study was purified by cold vacuum distillation before use. Well cleaned glass equipment was used, but no attempt was made to remove moisture/grease films or other trace impurities. All solvents were of reagent grade and were used as received. All work was carried out in ambient air, unless specified otherwise. The PPY-I, CT complexes were synthesized in aqueous dispersion as follows. One gram of solid iodine (sublimed, Aldrich Chemical Co.) was introduced into a 250 cm<sup>3</sup> beaker containing about 125 ml of deionized water. Since iodine is only sparingly soluble in water, most of the iodine remains as submerged solids. Then 1.5 ml of pyrrole was introduced onto the surface of the water. The beaker was covered with aluminium foil and kept in the dark either at room temperature or at 4°C for at least 48 h. As pyrrole diffused/dispersed slowly into the aqueous medium, the solution changed gradually from the original pink colour of  $I_2$  to dark green, characteristic of the CT complex formation, and eventually became almost colourless. A thick, black, spongy layer was obtained at the bottom of the beaker. The black spongy material was removed, rinsed thoroughly with copious amounts of methanol, acetonitrile, carbon tetrachloride and  $95\%$  ethanol (in that order). It was then soaked in a large excess of absolute ethanol for at least 24 h or boiled in absolute ethanol (78.5°C) for at least 1 h. The soaking or boiling

procedure was repeated until no unreacted monomer or acceptor could be detected in the ethanol wash by u.v. visible absorption spectroscopy. The black porous material was subsequently dried by pumping at a rotary vacuum of  $10^{-2}$  torr for at least 8 h. This black material was later identified to be the  $PPY-I_2$  CT complex (see text). The PPY- $I_2$  CT complex so produced is spongy and somewhat rubbery in texture. The yield was about  $35-40\%$ . Pyrrole was also polymerized in the presence of sulphuric acid initiator according to the method of Salmon<sup>16</sup>, except that  $0.5 g$  of iodine was also added to the reaction mixture in the present experiment. The polymer existed as a black swollen gel in the solution. The gel was then removed and washed thoroughly as described above for the preparation of the PPY- $I_2$  CT complex.

In the synthesis of PPY-Br,  $CT$  complex, 1 ml of bromine (BDH Chemicals, reagent grade) was dissolved in 50 ml of carbon tetrachloride in a  $250 \text{ cm}^3$  beaker. Then 75 ml of deionized water was added to form a twophase system. Monomer (1.5 ml) was introduced on to the top surface of the water layer. The system was covered with aluninium foil and allowed to stand in the dark for at least 60 h. A thick black polymer cake was obtained at the interface between the water and  $\text{CCI}_4$  layer. A thin brittle film (black) was also found floating on the water surface. Owing to its thin and brittle nature, the top film was not analysed further. The thick polymer cake was removed from the interface and subjected to a washing sequence as prescribed for the  $PPY-I$ , complex. It was then soaked in copious amounts of absolute ethanol for a prolonged period of time before being dried in dynamic vacuum. The yield was about  $45-50\%$ .

#### *Polymer characterization*

The u.v.-visible absorption spectra were measured using a Shimadzu UV-260 spectrophotometer equipped with a microprocessor. The i.r. absorption measurements were carried out on a Perkin-Elmer model 682 spectrophotometer with the polymer dispersed in KBr. SEM micrographs were obtained using a Philips SEM-505 scanning electron microscope. D.s.c. was performed using a Perkin-Elmer model DSC-2C calorimeter with a heating/cooling rate of 10 K min<sup>-1</sup> under a constant flow of nitrogen. T.g.a. was performed on a Netzsch model STA 409 simultaneous t.g.-d.t.a, apparatus at a heating rate  $10$  K min<sup>-1</sup> in nitrogen. The composition of the CT complexes was determined by elemental analysis. For electrical conductivity measurements, the polymer samples were pressed into thin circular pellets (about 0.05-0.1cm in thickness and 1.2cm in diameter) in a stainless-steel press at a pressure of about  $300 \text{ kg cm}^{-2}$ . Electrical conductivities were measured using the standard four-probe and two-probe techniques. Densities of the polymeric complexes at 25°C were determined by the flotation method in an ethanol/tetrabromoethane mixture. The density of the solution mixture was measured using a Paar model DMA 46 calculating digital density meter.

## RESULTS AND DISCUSSION

## *Morphology of the polymeric complexes*

*Figures 1* and 2 show the respective SEM micrographs of the PPY- $I_2$  and PPY-Br<sub>2</sub> CT complexes obtained



Figure 1 Scanning electron micrograph of  $PPY-I_2$  complex synthesized in aqueous dispersion



Figure 2 Scanning electron micrograph of PPY-Br<sub>2</sub> complex synthesized in  $H_2O/CCl_4$  two-phase solvent system

from CT polymerization in aqueous media. The  $PPY-I_2$ complex is highly porous and spongelike and is somewhat elastic in texture. Upon crushing and grinding, the spongelike structure disintegrates into granules. The PPY CT complex synthesized in the presence of sulphuric acid initiator according to the method of Salmon<sup>16</sup>, but with  $I_2$ also added to the reaction mixture (PPY-I<sub>2</sub>S), has only a granular morphology. As shown in *Figure 2,* the PPY-Br<sub>2</sub> CT complex is composed mainly of loosely packed granules/particulates of about 1 to several micrometres in dimension. The complex also lacks the rubbery texture of the  $PPY-I_2$  complex.

#### *U.v.-visible and infra-red absorption spectra*

The u.v.-visible absorption spectra of both the  $PPY-I_2$ and  $PPY-Br_2$  CT complexes reveal two relatively broad bands. One band centres at around 420nm and is characteristic of polypyrrole<sup> $6,16$ </sup>. The other intense broad

band which appears in the red and extends well into the near-i.r, region probably results from the CT interaction between the polymer and the dopant, since undoped PPY does not have any appreciable absorption in the neari.r.<sup>16</sup> The i.r. absorption spectra for the PPY-I<sub>2</sub> and PPY-Br 2 are shown in *Figures 3a* and *3b,* respectively, while that of the PPY- $I_2S$  is shown in *Figure 3c*. The i.r. spectra in all three cases show a featureless decrease in absorption from 4000 to about  $1700 \text{ cm}^{-1}$ . This long absorption tail has been assigned to be the tail of an absorption peak located in the near-i.r, for oxidized  $PPY^{6,18}$ . The absorption tail precludes the observation of the expected N-H stretching mode absorption between  $3300$  and  $3500 \text{cm}^{-1}$ . However, the bands at about 1540, 1300, 1170, 1030 and  $900 \text{ cm}^{-1}$  are all characteristic of  $PPY<sup>18</sup>$ . Comparison of the absorption spectra between  $PPY-I_2$  and  $PPY-Br_2$  reveals some differences in the relative absorbance of the respective PPY bands. This



Figure 3 Infra-red absorption spectra of (a)  $PPY-I_2$  complex, (b)  $PPY-Br_2$  complex and (c)  $PPY-I_2S$  complex

probably reflects the difference in the degree of CT interaction between the polymer and the dopant in the two complexes. A small band at around  $1700 \text{ cm}^{-1}$  can be clearly seen in PPY-Br<sub>2</sub> and PPY-I<sub>2</sub>S (but less so in  $PPY-I_2$ ) and is probably due to the presence of the pyrrolidinone carbonyl group, the presence of which has been reported for chemically synthesized PPY<sup>15</sup>. Finally, the bands at 1110 and  $610 \text{ cm}^{-1}$  for the PPY-I<sub>2</sub>S spectra (*Figure 3c*) are characteristic of the sulphate anion<sup>16</sup>.

## *Chemical compositions and electrical properties*

Results of the elemental analyses of the PPY-halogen CT complexes produced under various experimental conditions and their respective electrical conductivities are summarized in *Table 1.* The elemental analysis data indicate that the CT complexes are primarily polymerized pyrrole units plus the halogen dopant.

For the PPY-I<sub>2</sub> CT complex, there is no significant difference between the sample polymerized at room temperature and that at 4°C, as far as chemical compositions are concerned, although low-temperature polymerization appears to give a somewhat higher conductivity. Re-exposure of a well washed  $PPY-I_2$ complex to  $I<sub>2</sub>$  vapour can result in further uptake of the dopant (sample 3). However, no significant increase in conductivity was observed and the sample lost the excess iodine gradually when exposed again to the ambient air. Thus, the ratio of about four pyrrole units to one molecule of  $I_2$  in samples 1 and 2 may represent the stable, optimum stoichiometric composition for the PPY-I,  $CT$ complex. The result has been further confirmed by thermal analysis (see below). The ratio of about 3-4 pyrrole units to one anion has also been reported for a number of PPY-anion salt complexes obtained from electrochemical polymerization<sup>6,11</sup>. Polymerization of PPY in the simultaneous presence of  $H_2SO_4$  and  $I_2$ (sample 4) results in the uptake of about 0.18 moles of sulphate ion for every mole of pyrrole, a value comparable with that reported for the chemical polymerization of pyrrole by  $H_2SO_4$  alone<sup>16</sup>. However, the uptake of sulphate ion has greatly reduced the amount of  $I_2$  incorporation. As a result, a substantially lower conductivity than that of the  $PPY-I_2$  complex was observed for the PPY-I<sub>2</sub>S CT complex.

In the case of  $PPY-Br_2$  complex, the conductivity is lower than that of the  $PPY-I_2$  CT complex by about an order of magnitude. The stable, optimum stoichiometric composition for the  $PPY-Br_2$  complex corresponds to a pyrrole: $Br_2$  ratio of about 2.3:1. This pyrrole: $Br_2$  ratio is comparable to that found in the bromine-doped complex, using an originally insulating PPY film obtained from chemical polymerization in the presence of an acid

**Table 1** Chemical composition and electrical conductivity of the  $PPY-I_2$  and  $PPY-Br_2$  complexes

Sample no.	System	Polymerization medium	Chemical composition (normalized to $1.0 \text{ N}$ )	Conductivity $\sigma$ (ohm <sup>-1</sup> cm <sup>-1</sup> ) at 27 <sup>o</sup> C
	$PPY-I$	$H2O2$ , 27 <sup>°</sup> C	$C_{4.1}H_{2.6}N_{1.0}O_{1.0}(I_2)_{0.25}$	
	$PPY-I$	H <sub>2</sub> O, 4°C	$C_{4.1}H_{2.7}N_{1.0}O_{0.95}(I_2)_{0.26}$	
	$PPY-I2$	$H2O2$ , 27 $°C1$	$C_{4.0}H_{2.3}N_{1.0}O_{1.1}(I_2)_{0.44}$	
4	$PPY-I, S$	$H_2O/H_2SO_4$ , 27°C	$C_{4,2}H_{3,1}N_{1,0}O_{1,1}(SO_4)_{0,18}(I_2)_{0,02}$	$10^{-1}$
	$PPY-Br$ ,	$H_2O/CCl4, 27°C$	$C_{4,3}H_{3,1}N_{1,0}O_{1,2}(Br_2)_{0,44}$	$5 \times 10^{-1}$
6.	$PPY-I2$	$H2O2 27oC$		$3 \times 10^{-6}$

"Obtained by re-exposure of a well washed, stable PPY-I, CT complex to an  $I_2$  atmosphere  $P'PPY-I_2$  sample exposed to 5 M NH<sub>4</sub>OH solution for 36 h

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catalyst alone<sup>16</sup>. For comparison, the latter has a ratio of about 2.2 pyrrole units to one molecule of bromine.

The oxygen content in all samples was determined by difference. They are comparable to that of the PPY produced by the chemical oxidation of pyrrole<sup>16</sup>. This suggests, therefore, the presence of a substantial amount of pyrrolidinone structure<sup>15</sup> in the present polymeric complexes. Furthermore, we cannot exclude the possibility that the polymer samples have also been doped, to a small extent, by oxygen, as in the case of neutral and oxidized PPY films obtained from electrochemical polymerization<sup>6</sup>. The H mole ratio in the PPY-I, CT complex is substantially below the theoretical value of 3 to justify the primarily  $\alpha, \alpha'$  linkages of the pyrrole moiety<sup>6</sup>. The deficiency in H may serve to indicate the simultaneous presence of  $\alpha$  and  $\beta$  linkages for some of the pyrrole units of this complex, probably as a result of crosslinking or chain branching. However, the C:H:N ratio for the PPY-Br,  $CT$  complex is close to the expected ratio for a long linear chain of disubstituted pyrrole rings, i.e. 4:3:1.

We wish to emphasize that both the  $PPY-I_2$  and  $PPY-$ Br<sub>2</sub> CT complexes synthesized by the present method are stable in the atmosphere. The electrical conductivity in both cases shows only a weak temperature dependence, at temperatures above room temperature, and remains relatively stable up to about  $150^{\circ}$ C in the case of PPY-I, complex and about  $120^{\circ}$ C in the case of PPY-Br, complex (see below). However, a substantial loss in conductivity ( $> 10^6$ ) was observed when the PPY-I<sub>2</sub> CT complex was contacted with 5 M  $NH<sub>4</sub>OH$  solution for about 36 h (sample 6). This compensation behaviour is not unlike that of other conjugated polymers, such as polyacetylene<sup>19</sup> or polyphenylacetylene<sup>20</sup>, doped with electron acceptors.

The densities of the PPY-I<sub>2</sub> (sample 1) and PPY-Br<sub>2</sub> (sample 5) complexes produced via the present CT polymerization technique were determined, using the flotation method in ethanol/tetrabromoethane mixture, to be 1.99 g cm<sup>-3</sup> and 1.78 g cm<sup>-3</sup>, respectively.

#### *Thermal properties*

The d.s.c. scans of the PPY-I<sub>2</sub> and PPY-Br<sub>2</sub> CT complexes are shown in *Figure 4*. The scan for the PPY-I<sub>2</sub> CT complex does not show any obvious transition between 30 and 250°C, consistent with the scans reported for the PPY obtained from electrochemical polymerization<sup>18</sup>, except for the presence of an exothermic reaction, barely visible between 110 and 150°C, and a relatively sharp endothermic peak at about 183°C in the present sample. It is interesting to note that the peak at 183°C not only coincides with the boiling point of iodine (184°C) but also corresponds to the onset of major weight loss for the PPY-I<sub>2</sub> CT complex (see t.g.a. data below). Thus, the endothermic peak at 183°C may be associated with the physical vaporization or removal of  $I_2$  from the  $PPY-I$ , CT complex. The conclusion is further supported by the fact that in the second heating, after the sample is allowed to cool from 250°C to room temperature, the scan becomes featureless. The thermal response of the  $PPY-Br<sub>2</sub>$  CT complex is somewhat more complicated. The d.s.c. scan is featureless at the boiling point of  $\text{Br}_2$ (59°C), but shows a relatively broad exothermic peak centred at about 140°C and a superficial transition at about 197°C. The exothermic process is irreversible and



**Figure 4** D.s.c. scans of  $PPY-I_2$  (-complexes prepared at 27°C ----) and  $PPY-Br_2$  (---)



Figure 5 Thermogravimetric scans of PPY-I<sub>2</sub> (- $(--)$  complexes prepared at 27 $\mathrm{C}$  $-$ ) and PPY-Br<sub>2</sub>

does not recover when the sample is cooled. The temperature of this exothermic peak at about 140°C also corresponds to the onset of major weight loss in t.g.a. (see below). Thus, this exothermic process may be associated with the thermal decomposition reaction of the  $PPY-Br_2$ complex.

The thermogravimetric scans in nitrogen of the PPY- $I_2$  and PPY-Br<sub>2</sub> CT complexes from room-temperature polymerization are shown in *Figure 5.* Both complexes suffer only about  $65-70\%$  loss of their original weight when heated from room temperature to a temperature as high as 750°C. The PPY-I<sub>2</sub> complex is stable to heating up to slightly above the vaporization temperature of iodine (184°C), when it begins to lose weight gradually. For scans carried out in air, the complex also exhibits the same thermal stability below 184°C. Similarly, the PPY-Br, complex is thermally stable up to the decomposition temperature of about 140°C. In both cases, the weight losses are associated with a loss in electrical conductivity. The thermal stability of the present PPY-halogen CT complex differs significantly from the thermal behaviour of most of the halogen-doped conducting polymers, obtained in the conventional solid-gas phase interactions. Iodine-doped polyacetylene, for example,



**Figure 6** Thermogravimetric scans of  $PPY-I_2$  complex prepared at 4°C ( -), PPY-I<sub>2</sub> complex with excess  $I_2$  (---) and PPY-I<sub>2</sub>S complex  $(- \cdot - \cdot -)$ 

shows a rather poor thermal stability and suffers a weight loss immediately upon heating above room temperature<sup>21</sup>. The thermal instability of the polyacetyleneiodine complex may readily result from the morphological complexity of the pristine polymer and the chemical and physical heterogeneity usually associated with the vapour-phase doping<sup>22</sup>. Thus, the PPY-I<sub>2</sub> and  $PPY-Br<sub>2</sub>$  CT complexes obtained from the present CT polymerization technique must be of greater chemical and physical homogeneity, as a result of more uniform interaction between the dopant and the polymer.

For comparison, *Figure 6* shows the t.g.a. scans in nitrogen of the PPY-I<sub>2</sub> CT complex synthesized at  $4^{\circ}C$ , the PPY-I<sub>2</sub>S CT complex and the PPY-I<sub>2</sub> CT complex with 'excess' iodine obtained through further contact of a 'stoichiometric' PPY-I<sub>2</sub> complex with I<sub>2</sub> vapour (sample 3, Table 1). The thermal behaviour of the  $PPY-I_2$  CT complex prepared at 4°C is similar to that prepared at room temperature. However, for the PPY-I<sub>2</sub> sample with 'excess' iodine, the weight loss commences at a temperature as low as 115°C, with a more substantial loss occurring after 180°C. This observation readily indicates that for a PPY-I, CT complex with an  $I_2$ /pyrrole mole ratio greater than 0.25, the excess iodine molecules are only loosely complexed with the polymer. The  $PPY-I_2S$ complex loses initially about  $5\%$  of its weight when heated to above 60°C, followed by a major decomposition reaction at about 180°C. However, among all the complexes studied, it suffers the least total weight loss when exposed to high temperature. In fact, the  $PPY-I_2S$ complex still retains almost 50% of its original weight when heated to a temperature as high as 750°C. Finally, we wish to point out that the gradual weight loss observed among the present PPY-halogen CT complexes differs somewhat from that of PPY films obtained from electrochemical polymerization. The latter show a much more rapid weight loss when heated above the  $decomposition$  temperature<sup>18</sup>.

#### **CONCLUSIONS**

A relatively simple method has been found for the simultaneous chemical synthesis and doping of polypyrrole in aqueous media and in the presence of a halogenic electron acceptor, such as bromine or iodine. The lone pair of electrons on the pyrrole nitrogen might have made the molecule sufficiently electron donating to facilitate partial electron transfer to the halogenic electron acceptor, the pyrrole moiety becoming simultaneously sufficiently polarized for polymerization of the monomer to occur. Presumably the ionic nature of this interaction is not affected to a significant extent by the protic character of the aqueous media, since halogens are only sparingly soluble and solvation of the anions may be minimal. The polymer-dopant CT complexes so produced are of good thermal stability. The PPY-I, CT complex is spongelike and somewhat rubbery in texture and has a conductivity of the order of  $10<sup>1</sup>$  ohm<sup>-1</sup> cm<sup>-1</sup>. The PPY-Br<sub>2</sub> CT complex is granular in nature and has a conductivity of about  $0.5$  ohm<sup>-1</sup> cm<sup>-1</sup>. Both complexes show remarkable stability in the atmosphere. The same CT polymerization techniques are currently being explored for the synthesis of other conducting polyheterocyclics in our laboratory.

#### REFERENCES

- 1 Chiang, C. K., Fincher, C. R., Jr, Park, Y. K., Heeger, A. J., Shirakawa, H., Louis, E. J., Gau, S. C. and MacDiarmid, A. G. *Phys. Rev. Lett.* 1977, 39, 1098
- 2 Street, G. B. and Clarke, T. C. *IBM J. Res. Dev.* 1981, 25, 51<br>3 Robalt, J. F., Clarke, T. C., Kanazawa, K. K., Reynolds, J. R 3 Robalt, J. F., Clarke, T. C., Kanazawa, K, K., Reynolds, J. R.
- and Street, *G. B. J. Chem. Soc., Chem. Commun.* 1980, 347 4 Block, H. *Adv. Polym. Sci.* 1979, 33, 94
- 5 Kaneto, K., Yoshino, K. and Inuishi, Y. *Solid State Commun.*
- 1983, 46, 389 6 Street, G. B., Clarke, T. C., Krounbi, M., Kanazawa, K., Lee, V., Ptluger, P., Scott, J. C. and Weiser, G. *Mol. Cryst. Liz, Cryst.*
- 1982, 83, 253
- 7 De Young, H. G. *High Tech.* 1983, 3(1), 65 8 Diax, A. F. *Chem. Sci.* 1981, 17, 145
- 
- 9 Tourillon, G. and Garnier, *F. J. Electroanal. Chem.* 1982, 135, 173
- 10 Bargon, J., Mohmand, M. and Waltman, R. *IBM J. Res. Dev.*  1983, 27, 330
- 11 Kanazawa, K. K., Diaz, A. F., Geiss, R. H., Gill, W. D., Kwak, J. F., Logan, J. A., Rabolt, J. F. and Street, *G. B. J. Chem. Soc., Chem. Commun.* 1979, 854; Diaz, A. F., Kanazawa, K. and Gardini, *G. P. J. Chem. Soc., Chem. Commun.* 1979, 635
- 12 Diaz, A. F. and Castillo, *J. I. J. Chem. Soc., Chem. Commun.*  1980, 397
- 13 Diaz, A. F., Castillo, J. I., Logan, J. A. and Lee, W. Y. J. *Electroanal. Chem.* 1981, 129, 115
- 14 Gardini, G. P. *Adv. Heterocycl. Chem.* 1973, 15, 67
- 15 Jones, R. A. and Been, G. P. 'The Chemistry of Pyrroles', Academic Press, San Francisco, 1977
- 16 Salmon, M., Kanazawa, K. K., Diaz, A. F. and Krounbi, M. J. *Polym. Sci., Polym. Lett. Edn.* 1982, 20, 187; Nalwa, H. S., Dalton, L. R., Schmidt, W. F. and Rabe, J, G. *Polym. Commun.*  1985, 26, 240
- Bocchi, V. and Gardini, *G. P. J. Chem. Soc., Chem. Commun.*  1986, *148*  17
- De Paoli, M. A., Waltman, R. J., Diaz, A. F. and Bargon, J. J. *Polym. Sci., Polym. Chem. Edn.* 1985, 23, 1687 18
- Clarke, T. C. and Street, G. B. *Synthetic Metals* 1979/80, 119 19
- Kang, E. T. *Eur. Polym. J.* 1985, 21,919 20
- Osterholm, J. E., Yasuda, H. K. and Levenson, *L. L. J. Appl. Polym. Sci.* 1983, 28, 1265 21
- Tomkiewice, Y., Schultz, T. D., Brown, H. B., Clark, T. C. and Street, G, B. *Phys. Rev. B* 1981, 23, 5137 22